

# Relaxed GaP on Si with low threading dislocation density

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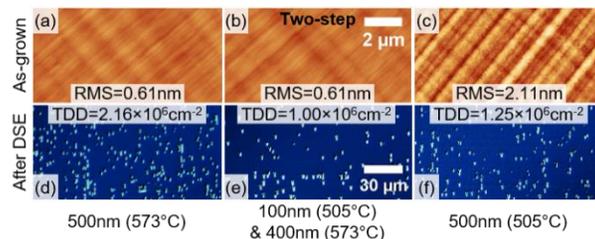
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Epitaxial growth of 1.7 eV GaAs<sub>0.75</sub>P<sub>0.25</sub> on GaP/Si templates offers a promising path to low-cost, high-efficiency tandem solar cells. While nucleation of thin, strained GaP on Si without anti-phase domains and stacking faults is now well established, the relaxation process of GaP on Si remains poorly understood. Threading dislocation densities (TDD)  $> 10^7$  cm<sup>-2</sup> in relaxed GaP on Si have been observed, despite the small lattice mismatch of  $\sim 0.45\%$ . Our prior work revealed that the TDD of relaxed GaP on Si is dominated by dislocation glide at low temperatures (e.g.  $< 505^\circ\text{C}$ ) and by dislocation nucleation at high temperatures (e.g.  $> 575^\circ\text{C}$ ), with lowest TDD of  $1.7 \times 10^6$  cm<sup>-2</sup> [1]. We also showed that the TDD of the relaxed GaP layer strongly influences the TDD and efficiency of the GaAs<sub>0.75</sub>P<sub>0.25</sub> active region, which emphasizes the importance of controlling the initial relaxation.

Here, we describe a two-step MBE growth process to suppress heterogeneous dislocation nucleation and improve dislocation glide during relaxation of GaP on Si. Initiating growth with a thin, low-T layer and subsequently growing a high-T layer for a combined thickness of  $\sim 0.5$   $\mu\text{m}$  enabled TDD reduction to  $1.0 \times 10^6$  cm<sup>-2</sup> in relaxed GaP on Si, which is the lowest value to date. We will show that the low-T step strongly suppresses dislocation nucleation, while the high-T step improves dislocation glide and surface morphology. This reduction in GaP TDD is expected to enable TDD values of  $2\text{-}3 \times 10^6$  cm<sup>-2</sup> for GaAs<sub>0.75</sub>P<sub>0.25</sub> solar cells on GaP/Si along with substantial efficiency improvements over current state-of-the-art.

Figure 1. Two-step growth improvement (center) over single-step growths. (a-c) AFM image (z-range = 0-17.8 nm) of as-grown surfaces. (d-f) Nomarski micrographs after defect-selective etching.



[1] K. N. Yaung et al., Appl. Phys. Lett. **109**, 032107 (2016).

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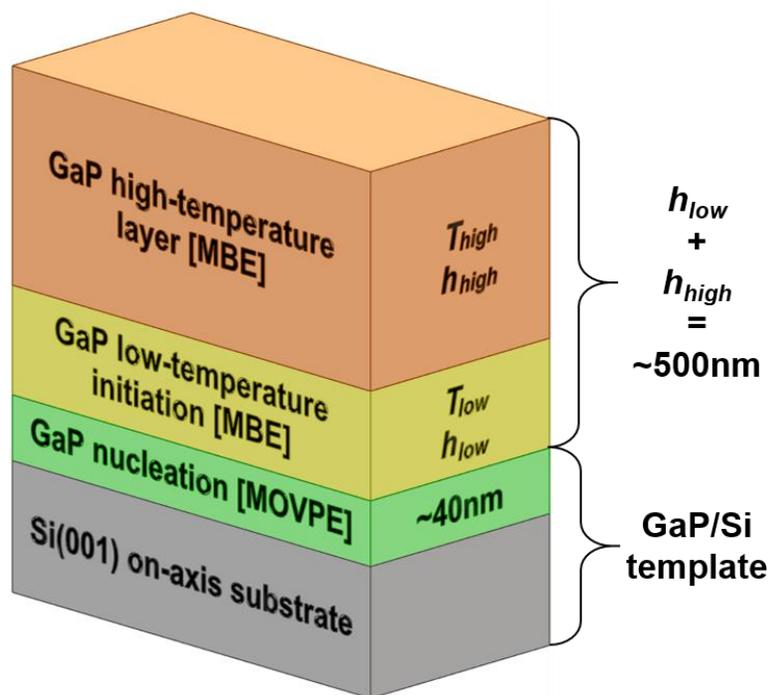


Figure 2. Layer structure of two-step GaP growth on GaP/Si templates. As-received GaP/Si templates have a ~40 nm GaP nucleation layer grown by metalorganic vapor phase epitaxy (MOVPE). Growth performed by solid-source molecular beam epitaxy (MBE): first a low-temperature,  $T_{low}$ , initiation layer of thickness,  $h_{low}$ , and then a high-temperature,  $T_{high}$ , layer of thickness,  $h_{high}$ . Total MBE-grown thickness is ~500 nm.

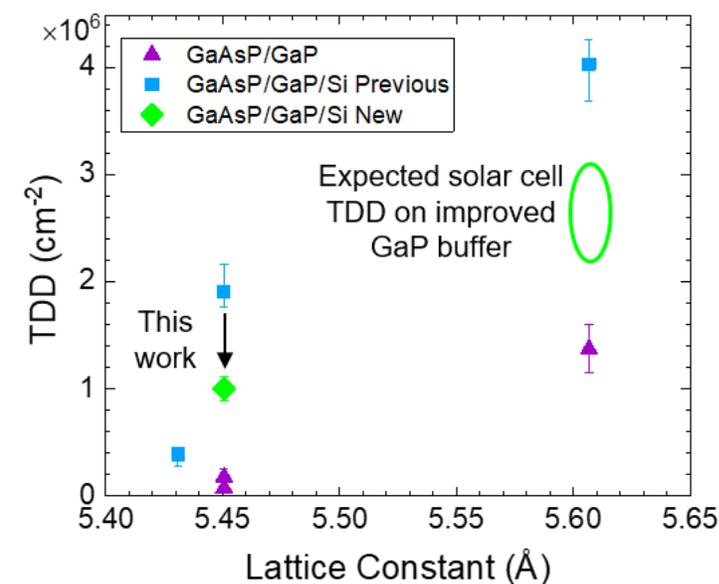


Figure 4. Evolution of TDD for growth on GaP/Si (blue, green) from template to final device as a function of relaxed lattice constant. Growth on bulk GaP (purple) also shown from substrate to device. Each is represented at the relaxed lattice constant (except for the nearly pseudomorphic GaP/Si template). TDD for GaP substrate and relaxed GaP determined by DSE; TDD for as-received GaP/Si template estimated using electron channeling contrast imaging. TDD of ~1.7 eV GaAsP (5.61 Å) determined by electron beam-induced current. Error bars show range of measurements from multiple images.

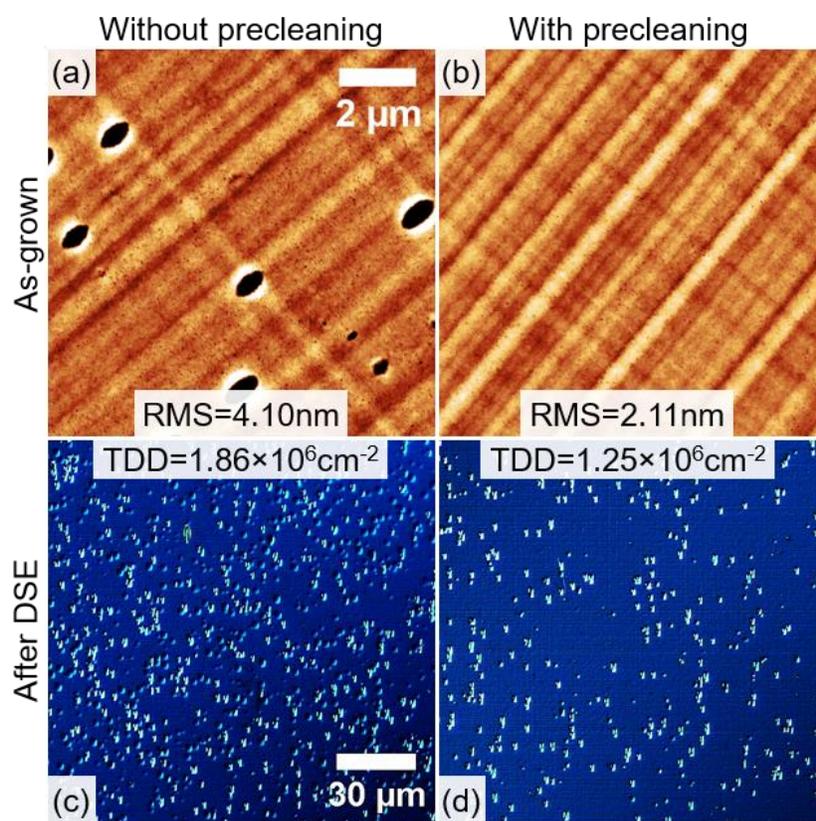


Figure 3. Comparison of GaP grown on (a,c) GaP/Si templates without precleaning and (b,d) templates with dilute aqua regia precleaning. Both samples were co-grown with single-step at 505°C. (a-b) Representative AFM images of as-grown surface morphology (z-range = 0-17.8 nm). Note pits for (a) extend lower than z-range with depth up to ~60 nm. (c-d) Representative Nomarski micrographs after defect selective etching (DSE). Precleaning of as-received GaP/Si templates lowered GaP TDD and improved as-grown surface morphology by reducing pitting, suggesting removal of heterogeneous dislocation nucleation sites.

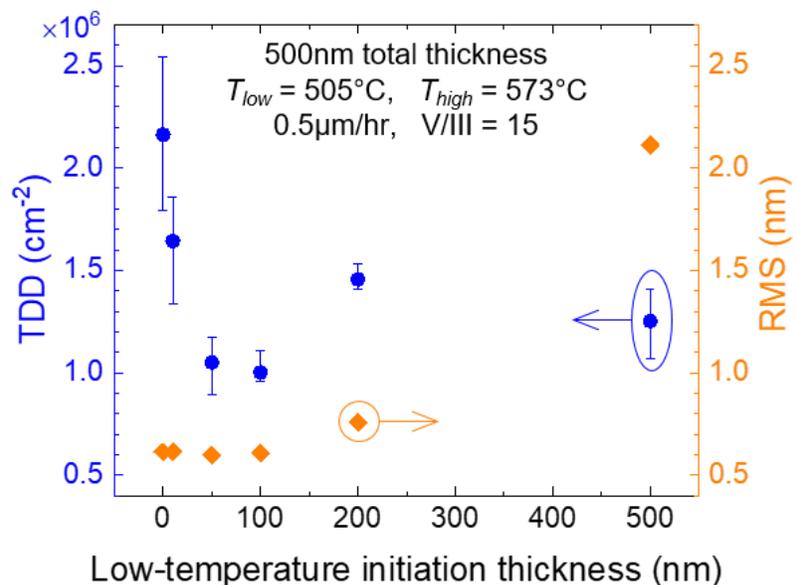


Figure 5. TDD and RMS roughness for relaxed GaP on GaP/Si as a function of low-temperature initiation layer thickness,  $h_{low}$ . Growth conditions are inset. All samples grown to ~500 nm of total MBE GaP thickness. Samples with  $h_{low} = 0$  nm and  $h_{low} = 500$  nm are single-step growths at 573°C and 505°C, respectively. RMS determined from  $10 \times 10 \mu\text{m}^2$  AFM images.

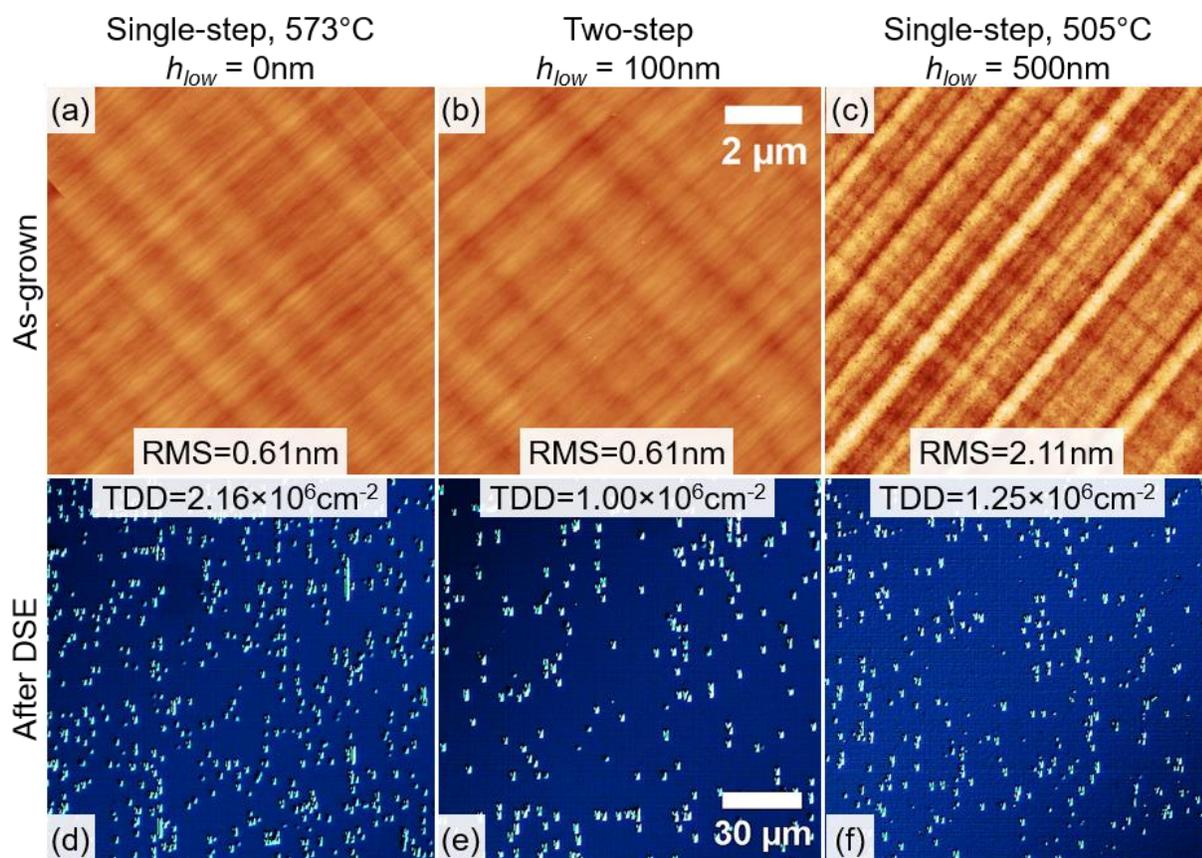


Figure 6. Expanded AFM images and Nomarski micrographs of Fig. 1 (main abstract page).

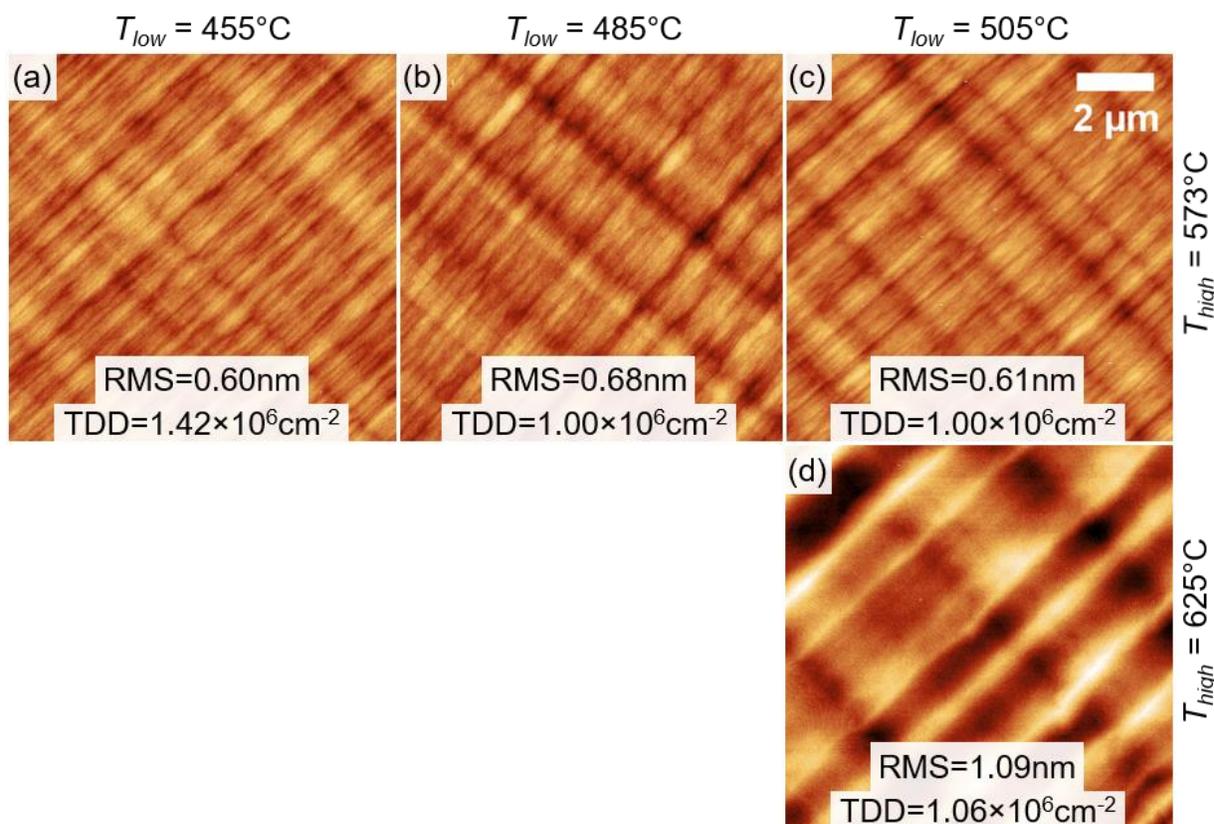


Figure 7. Comparison of surface morphologies for two-step growths with varied  $T_{low}$  and  $T_{high}$  (z-range = 0-6.33 nm). (a-c) Show varied  $T_{low}$  causes little surface morphology change, but increased TDD at  $T_{low} = 455^\circ\text{C}$ . (c-d) Show varied  $T_{high}$  results in little TDD change, but major surface morphology changes. Monolayer steps observed in (d) indicate change to step-flow growth for  $625^\circ\text{C}$  growth. In conventional single-step growth at  $625^\circ\text{C}$ , TDD of  $\sim 4 \times 10^6 \text{ cm}^{-2}$  is expected [1], which is  $\sim 4 \times$  higher than for two-step growth.